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ABSTRACT

Novel multi-functional materials with very low microwave (MW) absorbance in the X-band became demanding for varied high-sensitive electronic applications. To meet this goal, a new type of activated carbon samples containing fullerene-C_{70} was derived from coconut shells using the combined physical activation and milling process for the first time. The effects of various milling times (50, 75, and 100 minutes) on the structure, morphology, and MW reflection traits of these samples were examined. The crystalline phase of the activated fullerene-C_{70} was found to alter from cubic to rhombohedral structure at the milling time of 100, displaying the specific surface area of 36525 m^2/g and mean pore diameter of 3.42 nm. In addition, the permeability and permittivity of such activated carbon were improved with the increase of milling time. It was shown that by tuning the surface area and fullerene contents in the sample, the MW reflection loss of such activated carbon can be controlled. It is established that fullerene-C_{70} derived from the proposed activated carbon may be useful to produce low-cost and efficient MW absorption materials needed for diverse electronic devices with reduced electromagnetic interference.

Keywords: Activated carbon, Milling, Fullerene-C_{70}, Permeability, Permittivity, MW reflection loss

1. Introduction

With the development of microwave (MW)-based electronic technology the electromagnetic wave pollution became a serious environmental concern, greatly affecting human health and normal operations of the electronic devices. To mitigate these problems, electromagnetic wave absorbing materials have intensively been researched in recent years. Coating the target with an efficient electromagnetic wave absorbing material is a method to reduce the intensity of the reflected or transmitted electromagnetic waves [1–7]. This method utilizes the absorption or dispersion of electromagnetic energy in the medium between the electromagnetic wave source and the protected target. It imparts the electromagnetic wave absorbing materials the ability to absorb unwanted electromagnetic waves and stability against temperature and oxidation. In addition, they are easy to manufacture, flexible and affordable to become one of the *high-tech* essential materials.

Extensive researches revealed that the performance of the carbon nanomaterials and their composites like graphene, carbon spheres, carbon nanotubes are greatly potential in attenuating the electromagnetic waves [8–12]. Activated carbon is a carbon source that can be obtained by activating carbon physically or chemically. Surface morphology of the porous activated carbon is one of the unique characteristics that can be used in many applications such as dye absorbers, oil purification, supercapacitor electrodes, secondary battery electrodes, and others [8,13–17]. Recent progress show that the strategy of inserting magnetic components inside carbon materials is an effective way to improve the performance of the resultant products in absorbing electromagnetic radiation [18].

Considering the immense applied potential of the activated carbon materials, some activated carbon was derived from the coconut shell using the milling technique to achieve their improved MW absorption properties in the X-band. The milling times were varied to modify the structures, morphologies and electromagnetic radiation absorption attributes of the proposed activated carbon enclosing fullerene-C_{70}. A porous surface of the activated carbon with a very small volume was used to warp electromagnetic waves so that the internal surface reflections in the volume occurred, thus improving the heat dissipation quality of the electromagnetic energy. The as-prepared activated carbon samples were characterized to determine their milling time-dependent surface morphology, structure, specific surface area, porosity, and MW absorbance (in terms of reflection loss values) in the X-band.

2. Experimental

2.1. Preparation of activated carbon from coconut shells

Carbon was made via the carbonization process wherein coconut shells were burnt at 80°C for an hour under oxygen deficient conditions to eliminate the organic materials present in the shells. The loss of organic materials triggered the formation or opening of the carbon pores. Then, the burnt coconut shells-derived carbon was physically activated. Intense heat and water vapor enabled the severing of the carbon chains from the organic compounds. The heating as the physical activation process was intended to remove the impurities and impure hydrocarbons from the activated carbon. Next, the resultant carbon was heated in the temperature range of 800-900°C followed the water vapor streaming. The water vapor reacted with carbon, thus releasing carbon monoxide, carbon dioxide, and hydrogen. Thereafter, the activated carbon was pulverized using a

Mill Shaker at different milling times of 50, 75, and 100 minutes to obtain the particles' sizes in the micrometer range. The extracted activated carbon powders were labeled as C-50, C-75, and C-100 according to various milling times. A small amount of activated carbon was bound utilizing resin to form a rectangular shape steady with the WR90 sample holder for measuring the reflection loss (R_L).

2.2. *Characterizations*

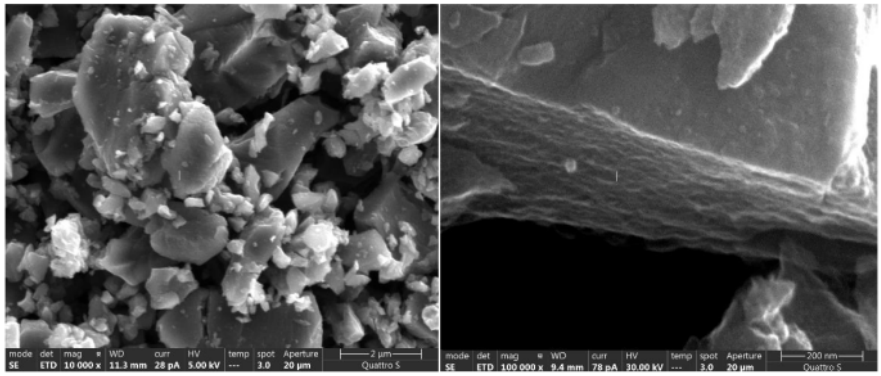
The morphology and microstructure of the prepared samples (C-50, C-75, and C-100) were examined using the field emission scanning electron microscope (FESEM, type Quattro S). The crystal structures and phases of the samples were measured using a SmartLab (3 kW) X-ray diffractometer equipped with the Cu–Kα line of wavelength (λ) ≈ 0.1541874 nm. The Surface Area Analyzer (SAA Quantachrome Instrument Version 11.03) was used to determine the specific surface area and pores diameter of the samples. The scattering characteristics (S) of the specimens were measured using a vector network analyzer (VNA) from Keysight (PNA-L N5232A). The MW absorbance values of the samples were calculated to get the components of S (S_{11}, S_{12}, S_{21}, and S_{22}). The values of S_{11} and S_{21} specifies the coefficients of reflection (Γ) and transmission (T), respectively. The recorded values of S_{22} and S_{12} were ignored due to their corresponding similarity with S_{11} and S_{21}. The values of complex relative permeability (μ_r) and permittivity (ε_r) were calculated using the Nicholson-Ross-Weir (NRW) method. In addition, the transmission/reflection line concept was used to calculate the reflection loss (R_L) of the samples, yielding the MW absorption characteristics.

3. Results and Discussion

3.1. *Morphology of the activated carbon*

Fig. 1 shows the SEM micrographs of C-50 sample at different magnification which consisted of irregular microstructures. The formation of carbon particles with inter-granular pores (Fig.1(a)) provided large specific surface area (Fig. 1 (b)) suitable of interacting strongly with the applied electromagnetic radiation. In sequence, these pores encouraged MW entrapment since the finer particles present in the pores could disperse the MW randomly in all directions, therefore altering the carbon's MW absorption traits. These unique forms of surface morphologies were

responsible for the differences in permeability, permittivity, and reflection loss characteristics in all activated carbon samples.



(a) (b)

Fig. 1. Morphology of C-50 at a magnification of (a) × 10000 and (b) × 100000

3.2. Structures and phase of the activated carbon

Fig. 2 displays the X-ray diffraction (XRD) patterns of all the milled samples. The XRD peaks of C-50 sample were matched to the cubic carbon (C – ICDD: 00-006-0675) with crystallographic parameters of $a = b = c = 0.35667$ nm and $\alpha = \beta = \gamma = 90°$. Similar observations were made for C-75 and C-100 samples wherein the Bragg's peaks were dominated by the cubic carbon lattice (C – ICDD: 00-006-0675). Sample grown at the milling times of 100 minutes (C-100) consisted of fullerene-C_{70} carbon phase (C- ICDD 00-048-1449), showing intense XRD peak at $2\theta = 23.90°$ with crystallographic parameters of $a = b = 0.98095$ nm, $c = 2.70220$ nm, and $\alpha=\beta= 90°$, $\gamma=120°$. It is known that fullerene-C_{70} enclosing seventy carbon atoms has spherical caged structure made of 25 hexagons and 12 pentagons connected by single and double covalent bonds. Due to this unique structure of fullerenes, strong internal reflections occur in the spherical cage, thus enabling it a gifted MW absorber. Earlier studies indicated that fullerene being one of the efficient reinforcing materials can be greatly effective in absorbing MW radiation[19]. In addition, it was claimed that the graphite – fullerene composites can exhibit excellent MW absorption performance compared to pure graphite[20] which needs further validation.

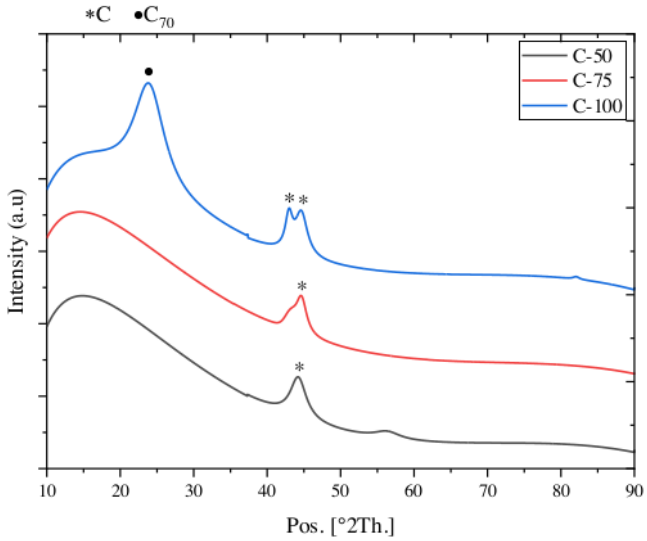


Fig. 2. XRD patterns of the as-prepared C-50, C-75, and C-100 samples with indicated carbon phases of ∗ C and • C70

3.3. *Specific surface area and pore diameter of the activated carbon*

Fig. 3 illustrates a multi-point isothermal BET plot of the activated carbon samples. Brunauer, Emmett, and Teller (BET) method was applied to evaluate the specific surface area (m^2/g) of the samples. These specific surface areas were used to determine the diffusion process through the porous material and selectivity for the catalyst reaction explained using the adsorption theory. The BET equation was used on the adsorption isotherms with P/P_0 values ranging from 0.05-0.3. The isothermal BET equation can be written as:

$$\frac{1}{W\left(\frac{P_0}{P}\right)-1} = \frac{1}{V_m C} + \left[\frac{C-1}{V_m C}\right]\frac{P}{P_0} \quad (1)$$

where W denotes the volume of absorbed gas at the relative pressure P/P_0, V_m is the volume of nitrogen gas that formed a monolayer at a solid surface, P is the pressure of adsorption equilibrium, P_0 is the pressure of adsorption saturation and C is constant of energy. The values of V_m for all the samples were estimated from the slope (s) and interceptions (i) on the BET chart, achieving the total and specific surface areas of the carbon as well as the average diameter of pores. The total

surface area and specific surface area of the C-50, C-75, and C-100 samples were discerned to be 1130, 345, and 947 m^2 and 50266, 13000, and 36525 m^2/g, respectively.

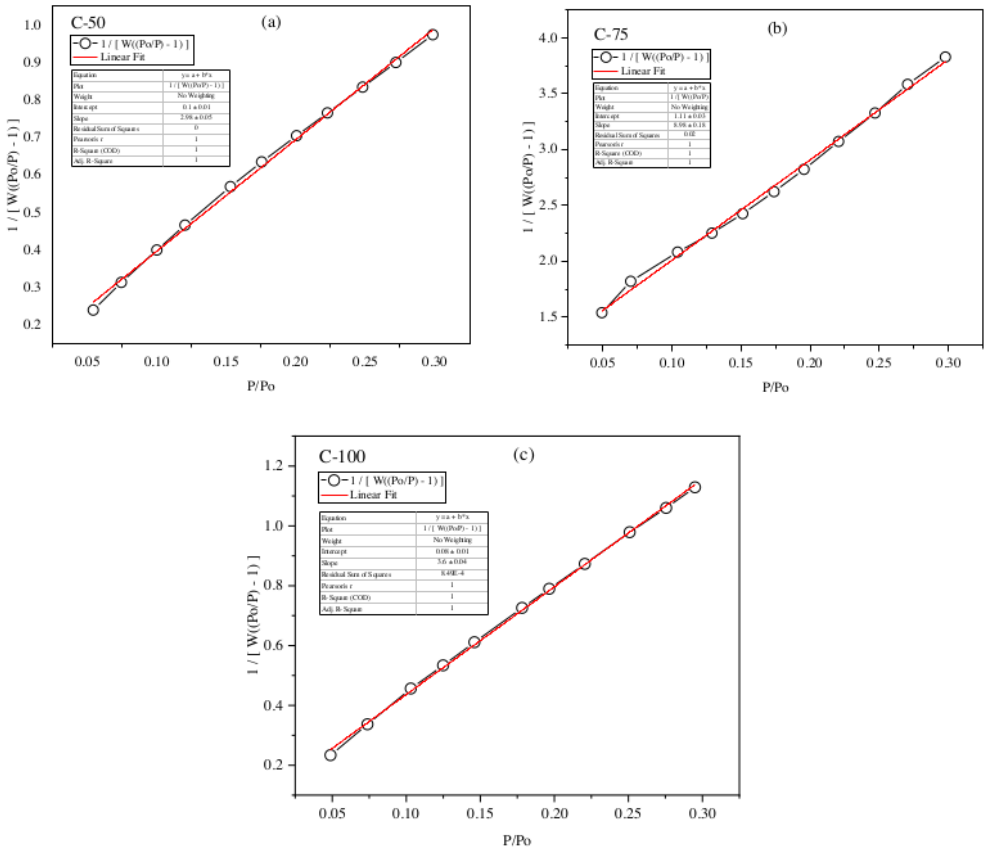


Fig. 3. Multi-point isothermal BET plot of the samples

Fig. 4 depicts the Barret Joyner Hallenda (BJH) pore size distribution - nitrogen adsorption at 77.35 K for C-50, C-75, and C-100 samples. Irrespective of the milling times, the pore radius distribution of the samples was in the range of 1 and 250 nm wherein the maximum peaks occurred at 1.53 (for C-50), 1.94 (for C-75), and 1.71 nm (for C-100). The average pore diameters of C-50, C-75, and C-100 sample were 3.06, 3.88, and 3.42 nm, respectively. According to International Union of Pure Applied Chemistry (IUPAC) standard, the resulting pore diameters were classified as the mesopores (pore diameter range of 2-50 nm). In addition, the pore volumes of C-50, C-75, and C-100 sample were 1.02×10^{-6}, 6×10^{-8}, and 7.4×10^{-7} m^3/g, respectively.

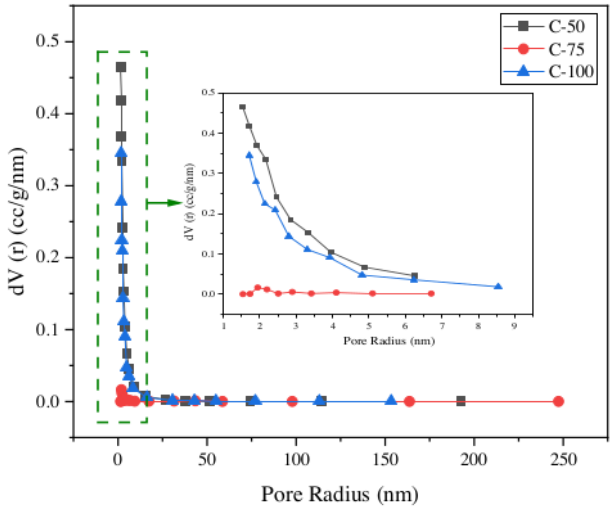


Fig. 4. BJH pore size distribution – nitrogen adsorption at 77.35 K for all activated carbon samples

Figs. 3 and 4 clearly revealed the activated carbon samples have large specific surface area and excellent porosity. This porous morphology play a significant role in the MW absorption, where a decrease in the pore diameter can appreciably increase the material's density or cavity concentration, thus leading to an alteration in the specific surface area responsible of the MW absorption. Consequently, an improvement in the specific surface area and porosity of the samples enables more interaction between carbon atoms and MW on the surfaces and interfaces.

2.3. Complex relative permeability and permittivity of the activated carbon

Fig. 5 displays the complex relative permeability and permittivity of C-50, C-75, and C-100 sample measured in the frequency range of 8.2-12.4 GHz. It is worth noting that the real permeability values of C-50, C-75, and C-100 samples were comparable due to their almost similar crystal phases. Conversely, C-100 sample showed higher magnetic energy storage capacity (higher value of real permeability) than C-50 and C-75 samples. Irrespective of the milling times, the real permeability values for all samples were gradually decreased to zero with the increase of MW frequency (Fig. 5(a)). The imaginary permeability of C-100 sample (Fig. 5(b)) was related to magnetic loss which dropped significantly to almost zero in the range of 8.2 to 9 GHz and then

fluctuated near zero. Meanwhile, the real permittivity of C-50 and C-75 samples (Fig. 5(c)) exhibited gradual declination trends with the increase of frequency. However, the real permittivity of C-100 sample remained steady with the increase of frequency, indicating an excellent electrical field energy storage performance of C-100. The real permittivity (or dielectric constant) also determines how much the incoming energy can be reflected and absorbed by the proposed activated carbon. In the frequency range of 8.2 and 9.3 GHz, the imaginary permittivity (or dielectric loss factor) of C-100 was considerably increased (Fig. 5(d)) that assessed the dissipation of electrical field energy in the form of heat inside the activated carbon.

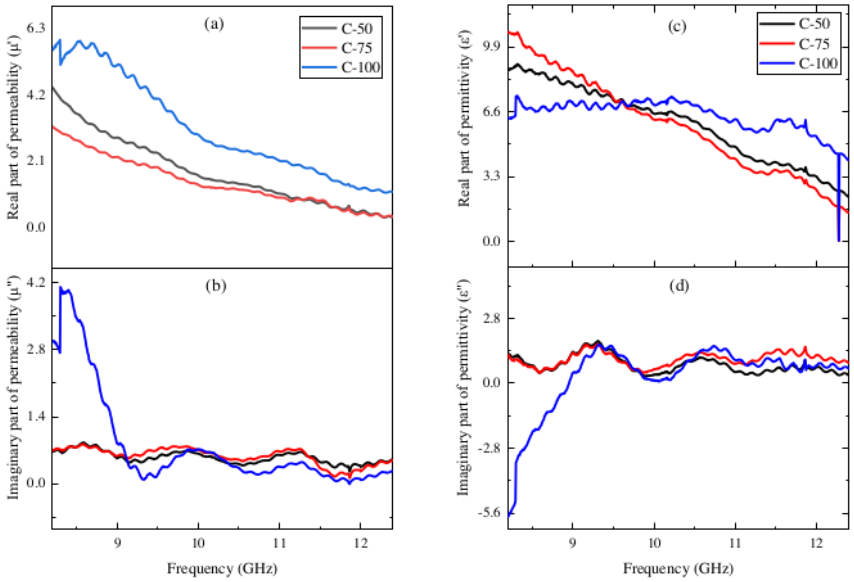


Fig. 5. MV frequency-dependent complex relative permeability and permittivity of 4 mm thick activated carbon.

2.4. Microwave reflection loss of activated carbon

Fig. 6 illustrates the MW absorption properties of all activated carbon specimens in terms of their frequency-dependent reflection loss (R_L). The R_L was found to be controlled by adjusting the milling time. The patterns of R_L for C-50 and C-75 were similar with an average value approximately -10 dB. Conversely, C-100 sample exhibited a prominent absorption band around 9 GHz with the bandwidth of 3 GHz. Interestingly, mesoporous C-100 sample containing

fullerene-C_{70} revealed significant MW absorbance in the X-band, indicating its potential as MW X-band applications in electronic devices. The produced fullerene-C_{70} dominated activated carbon also showed outstanding permeability and permittivity, suggesting its appropriateness for the MW X-band applications. Furthermore, the occurrence of a weak oscillation in the high-frequency range was attributed to the internal surface reflection within the mesopores. It was asserted that the milling process with optimal milling time could substantially enhance the microwave absorption capacity of the activated carbon. In short, the suggested fullerene-C70 enclosed activated carbon can lead to the development of low-cost, high-efficiency MW absorption materials desired for sundry applications.

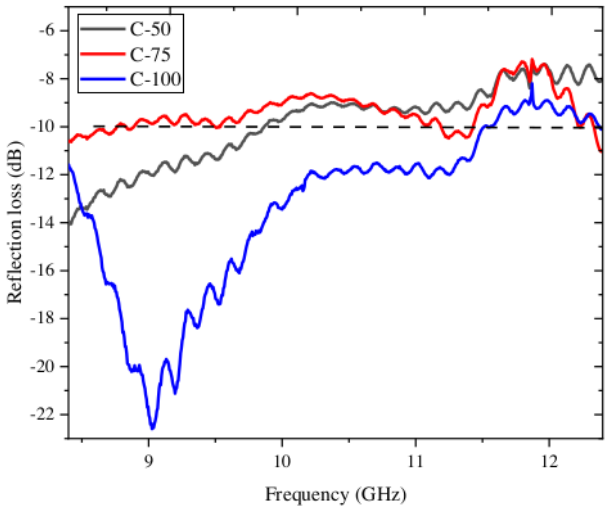


Fig. 6. MW reflection loss as a function of frequency for all activated carbon specimens

4. Conclusions

For the first time novel activated carbon specimens (C-50, C-75 and C-100) were prepared from the coconut shells using modified milling process plus heating process. The milling times were varied (50, 75, and 100 minutes) to get mesoporous activated carbon with improved MW absorption properties. The surface morphology, phase, structure, surface area, porosity, and MW reflection loss of these samples were improved with the increase of milling times. Activated carbon obtained as milling time of 100 minutes was the optimum one in terms of structures, morphology

and MW absorbance. The C-100 sample showed a phase transformation from cubic to rhombohedral crystal structure (or fullerene-C_{70}). The surface area and mean pore diameter of C-100 was correspondingly 36525 m^2/g and 3.42 nm. The activated carbon containing fullerene-C_{70} revealed excellent permeability and permittivity characteristics suitable for MW X-band applications. It was affirmed that by regulating the surface area and fullerene-C_{70} contents in the activated carbon the MW reflection loss can be tuned. The proposed fullerene-C_{70} based activated carbon can lead to the development of cheap and efficient MW absorption materials required for varied purposes.

Acknowledgments

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